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A Facile Method of Synthesis and Characterization of Sr Doped CeO₂ Nanocrystalline Materials

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Strontium doped Cerium oxide ($Ce_{1-0.25}Sr_{0.25}O_2$ and $Ce_{1-0.5}Sr_{0.5}O_2$) nanocrystals were synthesized by combustion technique. The synthesized ($Ce_{1-x}Sr_xO_2$, x = 0.25, 0.5) nanocrystals were annealed at 500° C for 2hr. The phase formation, surface morphology and elemental composition analysis have been studied using XRD, HRSEM and EDXS. XRD and HRSEM measurements revealed that the synthesized nanocrystals were in single phase with face-centered cubic (fcc) structure and that the nano-grains had different sizes. Chemical composition was confirmed using EDXS. The UV-Vis spectroscopic studies revealed that as the Sr doping level increases, the band gap width increases from $E_g = 2.9$ eV to 2.95 eV. At the same time, the wavelength of maximum absorbance shifts from $\lambda_{abs} = 285$ to 300 nm. Further, PL studies were performed with excitation wavelength $\lambda = 350$ nm and two main emission peaks were observed at 415 and 437 nm respectively. TGA, DTA and DSC measurements were carried out to check the weight loss and the occurrence of endothermic/exothermic reactions.

Keywords: Cerium oxide, Strontium metal, Combustion, XRD, HRSEM, EDX, UV, PL, TGA/DTA and DSC.

1 Introduction

Cerium (Ce), a rare earth metal belonging to the lanthanide series and one of the most abundant element in nature, exists in one of the two oxidation (reduced and oxidized) states: either cerous (Ce^{3+} , trivalent state) or ceric (Ce^{4+} , tetravalent state)¹. It is a ductile and malleable lustrous metal, which is iron-gray in color and highly reactive. Ceria (CeO₂) is a metal oxide n-type semiconductor², that has band gap energy $E_g = 3-3.6$ eV, high dielectric constant $\kappa = 23-26$, refractive index n: 2.2 - 2.8 and high dielectric strength up to 2.6 MV cm⁻¹. Cerium oxide (CeO₂) has exceptional technological importance due to its unique properties; particularly it behaves as a very good oxide-ion conductor in the case of solid oxide fuel cells and has excellent oxygen storage capacity, due to its high oxygen deficiency³⁻⁵. It is also used as an ultraviolet absorber ⁶⁻⁷, finds applications in gas sensors ⁸⁻¹³ while nanoceria plays a major role in cosmetic products as UV blockers ¹⁴.

In the recent years, doped CeO_2 has been an attractive candidate for both technological applications as well as scientific research. Various conventional techniques are available for the synthesis of Srontium

(Sr) doped CeO₂ nano-powders such as sol-gel¹⁵, hydrothermal¹⁶, thermal decomposition¹⁷, and combustion¹⁸. Among the various methods, the combustion technique is not only easier to handle but also efficient. Further, it is cost-effective and requires less effort to produce the ultrafine powder of Sr doped CeO₂ with narrow size distribution. In this work, nano-sized Sr doped Ceria ($Ce_{(1-x)}Sr_xO_2$, where x = 0.25, 0.5) have been synthesized by the combustion technique. We have characterized these samples and studied in detail the effect of doping CeO₂ with Sr. The results show that there is an enhancement of structural, optical, and thermal properties due to change in the dopant concentration.

2 Experimental Details

Commercially available powders of Cerium Nitrate [Ce (NO₃)₃.6H₂O], Urea [(NH₂)₂Co], Succinic acid [C₄H₆O₄], Strontium nitrate [Sr(NO₃)₂] (AR grade, Sigma Aldrich, USA, 99.9% purity) were used as starting materials. After synthesis, the doped samples Ce_{1-0.25}Sr_{0.25}O₂ and Ce_{1-0.5}Sr_{0.5}O₂ were annealed at a temperature of 500° C for 2 hr. XRD measurements were carried out for both the undoped and doped samples, using Bruker D8 Advance diffractometer with Cu_{Ka} radiation ($\lambda = 1.5406$ Å) as

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the radiation source, Ni filter and operating voltage of 40 kV at the current of 35 mA. The two doped samples were subjected to additional characterizations. Surface morphology and elemental analysis were studied using high resolution scanning electron microscopy and energy dispersive x-ray analysis (HRSEM with EDXS: Hitachi-S4800) at an operating voltage of 15kV. The UV spectra were measured with UV-Vis spectrophotometer (Shimadzu UV-2600) and the absorption spectrum was recorded in the wavelength range $\lambda = 200$ to 900 nm. The band gap energy was obtained by Kubelka-Munk function versus energy. The Photoluminescence (PL) measurement was carried out using HORIBA (FL3C-21-1962C-1318-FL) system with excitation wavelength $\lambda = 350$ nm. Thermal studies were carried out using simultaneous Thermogravimetric/Differential Thermal Analysis (TG/DTA), and Differential Scanning Calorimetry (DSC) in N2 atmosphere at a heating rate of 10°C/ minute. The recorded range of temperature profiles was calibrated with standard calibration materials (Nickel or iron, calcium (II) oxalate monohydrate CaC₂O₄.H₂O, silver and gold).

3 Results and Discussion

3.1 X-Ray Diffraction Studies

Fig.1 shows the powder X-ray diffraction (Cu_{ka}, $\lambda = 1.5406$ Å) of Sr doped CeO₂ sample. Fig.1(a)



Fig.1 — X-ray diffraction plots of Sr doped CeO₂ (a) Ce_(1-x)Sr_xO₂ (x=0) (b) Ce_(1-x)Sr_xO₂ (x = 0.25) annealed at 500° C for 2 hr, (c) Ce_(1-x)Sr_xO₂, (x = 0.5) annealed at 500° C for 2 hr.

shows the peaks of undoped CeO₂. In comparison Fig. 1(b) shows that the doped sample Ce_{1-0.25}Sr_{0.25}O₂ exhibits better defined peaks, *i.e.* higher crystalline quality after annealing. The diffraction peaks are indexed as (111), (200), (220), (311), (222), (400)^{*} and (331)^{*}corresponding to the diffraction angle $2\theta = 28.45^{\circ}$, 32.94° , 47.32° , 56.12° , 58.84° , 69.02° , and 76.49° , respectively. Stars indicate the presence of Sr peaks in the Cerium oxide matrix. It was noted that Sr addition actually increases the crystalline quality of the nano-powders, in particular for the Sr fraction 0.25 whereas with the fraction 0.50 the FWHM (full width at half maximum) of all significant peaks seems larger as seen in Fig 1(c).

The XRD patterns confirm the Fm 3 m space group, face-centered cubic structure with cell parameter (a = 5.43 Å \pm 0.0115) and volume =160.049 Å³. The average crystallite size was calculated by Scherrer's formula (JCPDS # 43-1002 for pure CeO₂, 65-5959 for Sr). The average crystallite size was found to be 2.63 nm for pure CeO₂ and 4.78 nm for the doped sample Ce_{1-0.25}Sr_{0.25}O₂. The Ce_{1-0.5}Sr_{0.5}O₂ doped sample on the other hand confirms the face-centered cubic structure but the average crystallite size was found to be 5.85 nm. Thus, one can conclude that increasing the dopant concentration increases the crystallite size with respect to that of pure Ceria.

3.2 HRSEM Spectroscopy

HRSEM morphology of Sr doped CeO₂ nanoparticles annealed at 500° C for 2 hr is shown in Fig. 2(a) Ce_{1-0.25}Sr_{0.25}O₂ and (b) Ce_{1-0.5}Sr_{0.5}O₂. It is observed that particles with porous nature as well as agglomerates are bonded together. The agglomeration was found to increase with increasing Sr doping of the CeO₂ samples.

3.3 EDXS Studies

The EDX spectra of the doped samples $Ce_{1-0.25}$ $Sr_{0.25}O_2$, and $Ce_{1-0.5}Sr_{0.5}O_2$ are shown in Figs. 3(a) and 3(b) respectively. The spectra clearly show the presence of the elements Sr, Ce and O, thus confirming the chemical composition of the powders.

3.4 UV-Visible Studies

Optical (UV-Visible) absorption spectra of Sr doped CeO₂ samples are shown in Fig. 4. For the samples Ce_{1-0.25}Sr_{0.25}O₂ and Ce_{1-0.5}Sr_{0.5}O₂, maximum absorption is found to occur at wavelengths $\lambda = 285$ nm and $\lambda = 300$ nm, respectively. The strong absorption peaks below 500 nm in the spectrum, is due to the charge transfer from O (2p) to Ce (4f) states in CeO₂



Fig. 2 — SEM images of Sr doped CeO₂: (a) $Ce_{1-0.25}Sr_{0.25}O_2$ and (b) $Ce_{1-0.5}Sr_{0.5}O_2$



Fig. 3 — EDX spectra of Sr doped CeO₂ nanoparticles: (a) $Ce_{1-0.25}$ Sr_{0.25}O₂, and (b) $Ce_{1-0.5}Sr_{0.5}O_2$ samples.

sample. The optical band gap was calculated from the Tauc plot of the Sr doped CeO_2 using the equation:

$$E_g = \frac{1240}{\lambda_{Absorp .Edge}}$$

Due to the increase in the Sr dopant in CeO₂, the optical properties also changes. That is, increase in Sr concentration in CeO₂ results in wider bandgap as well as increase of the wavelength at the maximum absorption. The band gap energies of Ce_{1-0.25}Sr_{0.25}O₂, and Ce_{1-0.5}Sr_{0.5}O₂ samples were calculated to be



Fig. 4 — Shows the optical absorption and optical band gap energy of $Ce_{1\cdot0.25}Sr_{0.25}O_2$ and $Ce_{1\cdot0.5}Sr_{0.5}O_2$ samples by Tauc plot.

 $E_g = 2.9 \text{ eV}$ and $E_g = 2.95 \text{ eV}$ respectively, using the Kubelka-Munk formula.

3.5 Photoluminescence

The photoluminescence of Sr doped CeO₂ (Ce_{1-0.25} Sr_{0.25}O₂ and Ce_{1-0.5}Sr_{0.5}O₂) nanoparticles was recorded at room temperature using Xenon lamp with excitation wavelength ($\lambda_{\text{excitation}} = 350$ nm). A broad emission from 400 to 500 nm was observed,

presenting two distinct features at 415 nm and 437 nm as seen in Fig. 5. In the case of Sr doped ceria $Ce_{1-0.25}Sr_{0.25}O_2$, the emission intensity of the peak around 415 nm was found to slightly shift towards higher wavelength upon Sr doping, which suggests the existence of a large amount of oxygen vacancies.

These oxygen vacancies facilitate the charge transfer or hopping between the Ce (4f) level and the O (2p) level (valence band). This also indicates that the surface defects increased due to the Sr doping.

3.6 Thermogravimetric Analysis (TGA)

Thermal analysis (TGA, DSC and DTA) is a group of techniques in which the physical/chemical properties of a substance and/or its reaction products are measured as a function of temperature, by subjecting the substance to a gradual temperature variation. Among these techniques, TGA was used to determine thermal stability (quantitative weight loss or decomposition) of the prepared $Ce_{1-0.25}Sr_{0.25}O_2$, and $Ce_{1-0.5}Sr_{0.5}O_2$ materials.

Fig. 6(a) shows two noticeable changes, first, in the temperature range from 50° C to 125° C, an initial weight loss (1.47%) was observed due to the evaporation of adsorbed water molecules and, second, in the temperature range from 125° C to 1000° C major weight loss (1.71%) was obtained, which is due to decomposition of residual organic moieties present in Sr doped CeO₂, (Ce_{1-0.25}Sr_{0.25}O₂) sample. The total weight loss was thus found to be 3.18% for Ce_{1-0.25}Sr_{0.25}O₂ and 6.43 % for Ce_{1-0.5}Sr_{0.5}O₂ as shown in Figs 6(a) and (b) respectively.



Fig. 5 — Photoluminescence spectrum of nanocrystalline $Ce_{1-0.25}$ $Sr_{0.25}O_2$, (black color) and $Ce_{1-0.5}Sr_{0.5}O_2$ (red color) samples.

3.7 DSC Studies

DSC is used to measure the amount of energy or power absorbed (endothermic) or released (exothermic) by the sample as it is heated/cooled or held at a constant temperature. In this study, the DSC curve appears as a single endothermic peak around 160.9° C for Ce_{1-0.25}Sr_{0.25}O₂ and 163.9° C for Ce_{1-0.5}Sr_{0.5}O₂ samples, which indicates an increase in enthalpy, however, it is yet to be understood if it involves a phase transition. (Fig.7).

3.8 DTA Studies

DTA measures the difference in temperature between the sample and reference material when both are heated/cooled at a uniform rate. This analysis was used to determine the endothermic or exothermic reactions of the prepared $Ce_{1-0.25}Sr_{0.25}O_2$, and $Ce_{1-0.5}Sr_{0.5}O_2$ materials and is shown in Fig. 8.

The DTA results show that an endothermic peak appeared around 125° C due to evaporation of water. The predominant exothermic peak around 800° C was obtained, due to decomposition of residual organic moieties present in Sr doped CeO₂.



Fig. 6 — Thermogravimetric analysis of (a) $Ce_{1-0.25}Sr_{0.25}O_2$ and (b) $Ce_{1-0.5}Sr_{0.5}O_2$ annealed at 500° C for 2h.



Fig. 7 — Differential scanning calorimetry of (a) $Ce_{1.0.25}Sr_{0.25}O_2$ and (b) $Ce_{1.0.5}Sr_{0.5}O_2$ at 500° C annealed for 2h.



Fig. 8 — Differential thermal analysis of (a) $Ce_{1-0.25}Sr_{0.25}O_2$ and (b) $Ce_{1-0.5}Sr_{0.5}O_2$ at 500° C annealed for 2h.

4. Conclusions

In this work, Sr doped CeO_2 nanocrystals, with Sr fractions of 0.25 and 0.50 were successfully synthesized by the combustion technique. After annealing at 500 °C the samples became crystalline and exhibited face-centered cubic structure as confirmed by XRD measurements.

The surface morphology and elemental compositional analysis was carried out by HRSEM and EDX on doped samples with different Sr content. The UV-Vis and PL measurements confirmed that as the Sr concentration increases in CeO_2 , the absorption peaks and band gap energy also shift to higher wavelength and higher energy respectively. The thermal analysis revealed that upon increase in the temperature, weight loss and endothermic or exothermic peaks were obtained.

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References

- 1 Korsvik C, Patil S, Seal S & Self W T, *Chem Commun*, 10 (2007) 1056.
- 2 Vangelista S, Piagge R, Ek S, Sarnet T, Ghidini G, Martella C & Lamperti A, *Thin Solid Films*, 636 (2017) 78.
- 3 Devaraju M K, Yin S & Sato T, ACS Appl Mater Interfaces, 1 (2009) 2694.
- 4 Djuricic B & Pickering S, J Eur Ceram Soc, 19 (1999) 1925.
- 5 Kamruddin M, Ajikumar P K, Nithya R, Tyagi A K & Raj B, *Scr Mater*, 50 (2004) 417.
- 6 Hernandes-Alonso M D, Hungria A B, Martinez- Arias A, Coronado J M, Conesa J C, Soria J & Fernandez-Garcia M, *Phys Chem Chem Phys*, 6 (2004) 3524.

- 7 Dao N N, Luu M D, Nguyen Q K & Kim B S, Adv Nat Sci Nanosci Nanotechnol, 2 (2011) 045013.
- 8 Zholobak N M, Ivanov V K, Shcherbakov A B, Shaporev A S, Polezhaeva O S, Baranchikov A Y, Spivak N Y, Tretyakov Y D, *J Photochem Photobiol B*, 102 (2011) 32.
- 9 Courbiere B, Auffan M, Rollais R., Tassistro V, Bonnefoy A, Botta A, Rose J, Orsiere T & Perrin J, *Int J Mol Sci*, 14 (2013) 21613.
- 10 De Marzi L, Monaco A, De Lapuente J, Ramos D, Borras M, Di Gioacchino M, Santucci S & Poma A, Int J Mol Sci, 14 (2013) 3065.
- 11 Demokritou P, Gass S, Pyrgiotakis G, Cohen J M, Goldsmith W, McKinney W, Frazer D, Ma J, Schwegler-Berry D, Brain J, & Castranova V, *Nanotoxicology*, 7 (2013) 1338.

- 12 Peng L, He X, Zhang P, Zhang J, Li Y, Zhang J, Ma Y, Ding Y, Wu Z, Chai Z & Zhang Z, *Int J Mol Sci*, 15 (2014) 6072.
- 13 Pulido-Reyes G, Rodea-Palomares I, Das S, Sakthivel T S, Leganes F, Rosal R, Seal S & Fernandez-Pinas F, *Sci Rep*, 5 (2015) 15613.
- 14 Zhang Y-W, Si R, Liao C-S, Yan, C-H, Xiao C-X & Kou Y, J Phys Chem B, 107 (2003) 10159.
- 15 Tambat S, Umale S & Sontakke S, Mater Res Bull, 76 (2016) 466.
- 16 Deus R C, Cortes J A, Ramirez M A, Ponce M A, Andres J, Rocha L S R, Longo E & Simoes A Z, *Mater Res Bull*, 70 (2015) 416.
- 17 Sharma J K, Srivastava P, Ameen S, Akhtar M S, Sengupta S K & Singh G, *Mater Res Bull*, 91 (2017) 98.
- 18 Umale S V, Tambat S N & Sontakke S M, *Mater Res Bull*, 94 (2017) 483.